



In-situ growing nickel phthalocyanine supramolecular structure on carbon nanotubes for efficient electrochemical CO₂ conversion

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ARTICLE INFO

Keywords:

Hydrogen-bonded organic ribbon
Nickel phthalocyanine
Carbon dioxide reduction reaction
Electrocatalysts

ABSTRACT

Heterogeneous catalysts made up of molecule-based catalysts on conductive substrates enable the large current density, high selectivity, and easy operation for carbon dioxide reduction reaction towards the practical use. Herein, we report a hydrogen-bonded supramolecular structure (**1**) assembled from asymmetric A₃B-type nickel phthalocyanine NiPc(OH)₆(DCNFO), which is afforded via solvothermal reaction of 2,3,9,10,16,17,23,24-octahydroxyl-phthalocyaninato nickel with tetrafluoroterephthalene. Single crystal X-ray diffraction analysis reveals the one-dimensional supramolecular ribbons of **1** formed from individual modules via hydrogen-bond between the hydroxyl groups, showing good solution resistance due mainly to the additional strong π - π interactions. Nevertheless, in-situ growing **1** on carbon nanotubes generates a phthalocyanine-based composite for gas diffusion electrode, displaying the highly selective conversion of CO₂ to CO with Faradaic efficiency > 98% at a wide voltage range from -0.5 to -1.2 V versus reversible hydrogen electrode, a high current density of 353 mA cm⁻² at -1.2 V, and stable cycling at 150 mA cm⁻² for 40 h in 0.5 M KHCO₃ electrolyte. Employment of 1.0 M KOH results in a further improved current density of 620 mA cm⁻² at -1.2 V and higher stable cycling current density of 230 mA cm⁻² for 40 h.

1. Introduction

The conversion of carbon dioxide to valuable fuels and chemicals is an important link for carbon circulation in a sustainable economy [1–9]. The electrochemical carbon dioxide reduction reaction (CO₂RR) has attracted increasing research interests due to the source renewability of electricity power as well as the mild operational conditions [10–14]. In this direction, inorganic catalysts and metal complexes are two kinds of dominant electrochemical CO₂RR catalysts. The former species with fast electron transport nature and rich active sites are capable of delivering high current densities to electrochemically convert CO₂ molecules [10–12]. Compared with inorganic electrocatalysts, homogeneous metal complexes are particularly important systems for precious mechanistic studies at the molecular level [12,15–23], ensuring the superior selectivity and specific activity in the CO₂RR by tailoring the well-defined active site structures. Heterogenizing molecular catalysts is generally preferable for practical applications, as demonstrated by the recent research progress in immobilizing molecules on conductive supports to

realize prominent catalytic performance [17–24]. Such heterogenization strategy enables the efficient electron transport to the catalytically active sites, dense spatial density of molecular complexes on electrodes, and operational convenience as well as improvement of stability and reusability. However, it is still an outstanding challenge to generate high current densities with high selectivity for molecule-based catalysts due possibly to their aggregation and low loading on conductive substrates.

Transition-metal tetrapyrroles such as metal porphyrins (MPor) and phthalocyanine (MPc) have been a kind of well-known metal complexes electrocatalysts for CO₂RR [25–29], with the first study of cobalt phthalocyanine (CoPc) for electrocatalytic CO₂ conversion initiated as early as in 1970s [30]. In the recent several years, MPor- and MPc-based electrocatalysts made up of molecular materials immobilized on conductive carbonaceous supports have sparked the renewed interests, because their superior heterogeneous electrocatalytic performances in water phase are comparable to some of the excellent metallic catalysts and metal-nitrogen-carbons [20,31–54]. In this regard, the excellent electrocatalytic activities of molecular metal phthalocyanines seriously

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<https://doi.org/10.1016/j.apcatb.2023.122446>

Received 2 December 2022; Received in revised form 16 January 2023; Accepted 8 February 2023

Available online 14 February 2023

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depend on their well-dispersed form on conductive substances as well as the innovation of electrolyzer with fast gas diffusion instead the liquid mode in H-type cell [20,33,39]. For example, the CoPc molecules dispersed on carbon nanotubes (CNTs) have been demonstrated to exhibit high Faradaic efficiency (FE) and turnover frequency (TOF), but usually at low current densities $< 40 \text{ mA cm}^{-2}$ towards electrochemical CO_2 conversion using H-type cell [55–58]. In addition, the high cycling current densities over 150 mA cm^{-2} accompanying with favorable selectivity for CoPc molecular catalyst have been achieved by using a flow cell [39]. Until recently, the discovery of nickel phthalocyanine compound in a molecularly dispersed form on CNTs, displaying $> 99.5\%$ CO_2 -to-CO selectivity at a high current density of 300 mA cm^{-2} , has distracted the research attention from CoPc catalysts to other metallic analogs [59–70]. It is worth noting that porous reticular materials provide another effective approach for molecular electrocatalysts heterogeneity by integrating catalytically active modules into metal–organic frameworks (MOFs) [3,36,37,48,71–73] and covalent organic frameworks (COFs) [6,42–47,74–78] with the help of coordination and covalent interactions, respectively. In contrast to molecular electrocatalysts, these molecule-based polymeric materials conventionally display the enhanced stability and high selectivity but still inferior current density due possibly to the inhomogeneous dispersion on conductive substances of grinding electrodes.

Herein, we introduce *in-situ* synthesis of nickel phthalocyanine supramolecular structure for **1** on CNTs to provide a **1**@CNT composite based on the solvothermal reaction of 2,3,9,10,16,17,23,24-octahydroxyl-phthalocyaninato nickel ($\text{NiPc}(\text{OH})_8$) with tetrafluoroterephthalene (TFPN) in N,N-dimethylformamide (DMF). This supramolecular structure of **1** is assembled from an asymmetric A_3B -type nickel phthalocyanine molecule $\text{NiPc}(\text{OH})_6(\text{DCNFO})$ through hydrogen-bond and π - π interactions, exhibiting insolubility and good solvent resistance. It therefore could be regarded as a supramolecular polymeric catalyst, different from those reported mononuclear metallic phthalocyanines molecules. **1**@CNT has been used as the gas diffusion electrode (GDE) towards CO_2 RR in 0.5 M KHCO_3 electrolyte, showing outstanding CO_2 -to-CO selectivity, as demonstrated by $\text{FE}_{\text{CO}} > 98\%$ at a wide voltage range from -0.5 to -1.2 V versus reversible hydrogen electrode (RHE). Moreover, the current density is able to reach 353 mA cm^{-2} at -1.2 V , and the stable current density of 150 mA cm^{-2} has been cycled for 40 h. In particular, employment of 1.0 M KOH electrolyte leads to the current density up to 620 mA cm^{-2} at -1.2 V as well as higher stable cycling current density of 230 mA cm^{-2} for 40 h. Theoretical calculations and experimental data disclose the increased electron density around nickel ion in $\text{NiPc}(\text{OH})_6(\text{DCNFO})$ in comparison with that of $\text{NiPc}(\text{OH})_8$ due to introduction of 3,6-dicyano-4,5-difluoro-1,2-phenylene-bisoxo group to the former compound. This, in combination with the thin-layered dispersion of **1** on CNTs, plays an important role in its excellent electrocatalytic performance.

2. Experimental

2.1. General remarks

All reagents were commercially available and directly employed. The synthesis of $\text{NiPc}(\text{OH})_8$ was referred to the previous literature procedure [71].

2.2. Preparation of **1**

A Pyrex tube ($9 \times 6 \text{ mm}$, o.d. \times i.d.) was filled with $\text{NiPc}(\text{OH})_8$ (14.0 mg, 0.02 mmol), TFPN (8.0 mg, 0.04 mmol), and DMF (2.0 mL). After sonicating the tube for 15 min followed by flash freezing in a liquid N_2 bath, the frozen tube was degassed and slowly thawed to room temperature. Such an operation was repeated for three times, then the tube was sealed by flame with a length of ca. 18 cm. The reaction was performed at 120°C for 3 days. The solid was isolated via centrifugation

and successively washed with DMF and acetone for three times. The product was collected and dried under air overnight to provide black-green needle-like single crystals of **1** (8.5 mg) in a yield of 36.9%.

2.3. Preparation of **1**@CNT(*n*)

1@CNT(22.1) was selected as representative to illustrate the synthesis of these composites. A Pyrex tube ($9 \times 6 \text{ mm}$, o.d. \times i.d.) was filled with $\text{NiPc}(\text{OH})_8$ (14.0 mg, 0.02 mmol), tetrafluoroterephthalene (8.0 mg, 0.04 mmol), carbon nanotubes (CNTs, 30 mg), and DMF (2.0 mL). After sonicating the tube for 15 min followed by flash freezing in a liquid N_2 bath, the frozen tube was degassed and slowly thawed to room temperature. Such an operation was repeated for three times, the tube was sealed by flame with a length of ca. 18 cm. The reaction was performed at 120°C for 3 days. The solid was isolated via centrifugation and successively washed with DMF and acetone for three times. The product was collected and dried at room temperature to give **1**@CNT(22.1). Following the same procedure, **1**@CNT(7.8), **1**@CNT(14.5), **1**@CNT(29.8), **1**@CNT(50.0), and **1**@CNT(75.0) were prepared by adding different amounts of CNTs, namely 100.0, 50.0, 20.0, 8.5, and 2.8 mg, respectively.

2.4. Preparation of **1**@CB

A mixture of **1** (2.0 mg) and carbon black (CB, 7.1 mg) was grinded to obtain **1**@CB.

2.5. Preparation of **1**@CNT

A mixture of **1** (2.0 mg) and CNTs (7.1 mg) was grinded to obtain **1**@CNT.

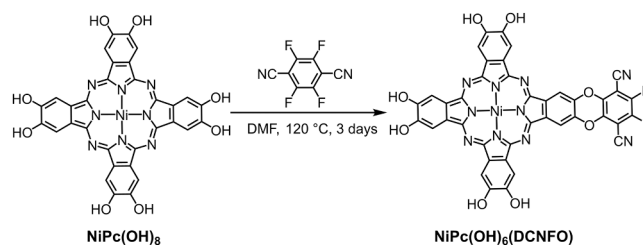
2.6. Preparation of $\text{NiPc}(\text{OH})_8$ @CNT

A Pyrex tube ($9 \times 6 \text{ mm}$, o.d. \times i.d.) was filled with $\text{NiPc}(\text{OH})_8$ (14.0 mg, 0.02 mmol), CNTs (30.0 mg), and DMF (2.0 mL). After sonicating the tube for 15 min followed by flash freezing in a liquid N_2 bath, the frozen tube was degassed and slowly thawed to room temperature. After such a process was repeated for three times, the tube was sealed by flame with a length of ca. 18 cm. The reaction was performed at 120°C for 3 days. The solid was isolated via centrifugation and successively washed with DMF and acetone three times. The product was collected and dried at room temperature overnight to $\text{NiPc}(\text{OH})_8$ @CNT as a control electrocatalyst.

3. Results and discussion

3.1. Crystal structure of **1**

A new nickel phthalocyanine compound **1** was prepared through the solvothermal reaction of $\text{NiPc}(\text{OH})_8$ with tetrafluoroterephthalene in DMF at 120°C for 72 h (Scheme 1). After the reaction, black-green needle-like crystals of **1** were observed, and the structure of this



Scheme 1. Synthesis of $\text{NiPc}(\text{OH})_6(\text{DCNFO})$ as building block for supramolecular compound **1**.

compound was solved by single crystal X-ray diffraction study. **1** subordinates to monoclinic system with space group $P2_1/n$ (Table S1). The definite structure of this compound is shown in Fig. 1a–c. Its molecular building block, $\text{NiPc}(\text{OH})_6(\text{DCNFO})$, is made up of a 3,6-dicyano-4,5-difluoro-1,2-phenylene-bisoxo group connecting on the 2,3-positions of 9,10,16,17,23,24-hexahydroxyl-phthalocyaninato nickel segment (Fig. 1a), forming a new asymmetric A_3B -type nickel phthalocyanine molecule with C_{2v} symmetry. The center nickel ion is coordinated with four isoindoles nitrogen atoms, and the Ni–N bond lengths are in the range of 1.896(3)–1.905(3) Å, which are comparable to the previous results of nickel phthalocyanine [79]. Each $\text{NiPc}(\text{OH})_6(\text{DCNFO})$ molecule interacts with four DMF molecules through the hydrogen bonds between four different hydroxyl groups and amide oxygen atoms of DMF molecules ($\text{O1} \cdots \text{H1} \cdots \text{O12}\#1$, $\text{O3} \cdots \text{H3A} \cdots \text{O9}$, $\text{O5} \cdots \text{H5} \cdots \text{O10}$ and $\text{O6} \cdots \text{H6A} \cdots \text{O11}$, symmetry code #1: $-x, 1-y, 1-z$) (Fig. 1a and Table S2). In addition, a $\text{NiPc}(\text{OH})_6(\text{DCNFO})$ molecule is also linked by two neighboring phthalocyanine molecules via hydrogen bonds among hydroxyl groups ($\text{O2} \cdots \text{H2} \cdots \text{O3}\#2$ and $\text{O4} \cdots \text{H4} \cdots \text{O2}\#3$, symmetry code #2: $-1/2 + x, 3/2 - y, 1/2 + z$ and #3: $1/2 + x, 3/2 - y, -1/2 + z$), forming a zigzag supramolecular ribbons (Fig. 1b and Table S2). These zigzag supramolecular chains are packed through strong π – π interactions, as indicated by the short distance of 3.257 Å and 3.331 Å between neighboring macrocyclic N_4 planes (Fig. 1c).

3.2. Characterizations of bulk **1**

The newly obtained compound **1** together with its two precursors, TFPN and $\text{NiPc}(\text{OH})_8$, were characterized by Fourier transform infrared (FT-IR) spectroscopy (Fig. 1d). **1** has the characteristic absorption peak of cyano ($\text{C}\equiv\text{N}$) from TFPN at 2244 cm^{-1} and the signals of ether ($\text{C}-\text{O}-\text{C}$) at 1277 cm^{-1} and 1010 cm^{-1} [46], disclosing the covalent connection of two TFPN and $\text{NiPc}(\text{OH})_8$ moieties. The UV–vis spectrum of $\text{NiPc}(\text{OH})_8$ in DMF exhibits a strong Q band at 685 nm, Fig. 1e. In contrast, the Q band of **1** dispersed in DMF has a red shift to 695 nm with a broad adsorption, indicating the presence of J -aggregation form in this compound due to the strong intermolecular π – π interactions. This is consistent with the observed slide angles of 26.0° and 42.3° between different phthalocyanine dimers in the crystal structure of **1**, (Fig. S1). Furthermore, elemental analysis of **1** is slight deviated from the chemical composition made up of a phthalocyanine molecule with four crystalized DMF molecules due to the partial solvent loss at room temperature (Table S3). The bulk solid was performed on powder X-ray diffractometer, and the diffraction peaks are overall similar to the simulated data from the single crystal structure (Fig. 1f). Pawley refinement upon powder X-ray diffraction (PXRD) data displays $a = 7.3622(3)\text{ Å}$, $b = 33.0215(11)\text{ Å}$, $c = 20.8136(6)\text{ Å}$, $\alpha = \gamma = 90^\circ$, and $\beta = 97.386(3)^\circ$ with $R_p = 4.34\%$ and $R_{wp} = 5.49\%$. This, in combination with the elemental analytic results, confirms the bulk product purity. This compound is insoluble in common organic solvents including acetone, dichloromethane, methanol, and tetrahydrofuran, and has a tiny

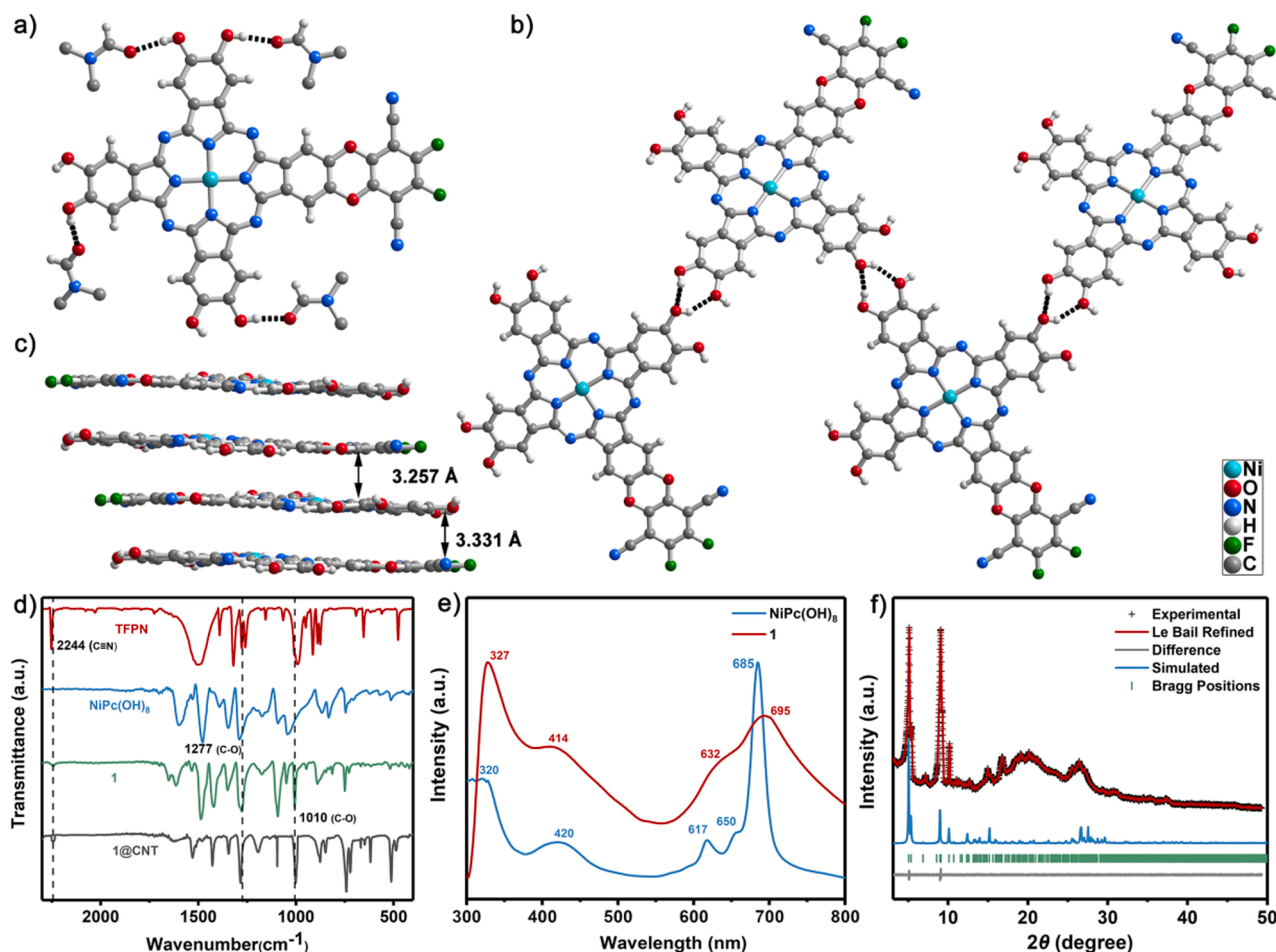


Fig. 1. (a) Crystal structure of an A_3B -type asymmetric $\text{NiPc}(\text{OH})_6(\text{DCNFO})$ molecule and neighboring four DMF molecules with hydrogen bonding connection in **1**; (b) Hydrogen bonding interactions between adjacent $\text{NiPc}(\text{OH})_6(\text{DCNFO})$ molecules in **1**; (c) π – π interactions in **1**; (d) Comparison in FT-IR spectra of **1**, $\text{NiPc}(\text{OH})_8$, TFPN, and **1**@CNT; (e) Comparison in UV–vis absorption of $\text{NiPc}(\text{OH})_8$ and **1**; (f) PXRD characterization of **1**: experimental PXRD profile (black), refined profile (red), difference (gray), and simulation pattern (cyan).

solubility in DMF (Fig. S2). After soaking **1** in organic solvents for one day, the unchanged crystal structure was confirmed by the consistent PXRD patterns with that of as-prepared species (Fig. S2), indicating the fact that this supramolecular structure possesses good solvent resistance, comparable with MOFs but different from molecular phthalocyanines. As a result, **1** should be a good heterogeneous catalyst for useful reactions. In addition, **1** shows good thermostability with the decomposed temperature higher than 400 °C (Fig. S3).

3.3. In-situ preparation and characterization of 1@CNT

The aggregating and poor conducting nature of molecule-based catalysts, whatever for metal complexes and polymeric materials, requires the necessary immobilization with conductive substrates. CNTs have unique electrical conductivity, high thermal stability and intrinsic mobility, as well as a high surface area that provides large specific surface area and rapid electron transfer. It can anchor molecular catalyst through physical adsorption and other forces [24]. As a result, **1** was doped on multiwalled CNTs using an in-situ preparation mode under the above-mentioned solvothermal reaction, giving the composites **1**@CNT (*n*) (*n* wt% = 7.8, 14.5, 22.1, and 29.8 wt% for the mass percentage of **1** in the composites, respectively). **1**@CNT(22.1) as a representative of this series of composites was carefully characterized. The FT-IR spectrum of **1**@CNT(22.1) displays the bands at 2244, 1319, and 1010 cm⁻¹ (Fig. 1d), indicating the attendance of **1** in this composite. On the other hand, the Q band of **1**@CNT(22.1) was slightly shifted to 699 nm as compared with that of the pristine NiPc(OH)₆(DCNFO) (695 nm) (Fig. S4), suggesting the presence of the interface interaction between NiPc(OH)₆(DCNFO) and CNTs. Nevertheless, the high resolution mass spectrometry (HRMS) data show only two ionic fragment *m/z* signals at 671.09 and 856.95 (Fig. S5), exactly corresponding to the molecular formula [M - 2 H]⁺ and [M - C₈F₂N₂O₂ + 6 H]⁺ (M = 859.3) for NiPc(OH)₆(DCNFO). This evidence indicates the in-situ immobilization of **1** component on CNTs (Fig. S5).

One-dimensional fiber-like morphology of **1**@CNT(22.1) was clearly captured by scanning electron microscope (SEM) photo, Fig. S6

(Supplementary Material). In contrast to the naked CNTs, the thickened diameter of **1**@CNT(22.1) shown in transmission electron microscope (TEM) photo suggests the successful immobilization by addition of CNTs into the solvothermal reaction (Fig. 2a and b). In addition, TEM and aberration corrected high angle annular dark-field scanning transmission electron microscopy (AC HAADF-STEM) photos clearly disclose that CNTs (in dark color) were intimately covered by thin-layered **1** with the thickness of 2–4 nm (Figs. 2b, c and S7). **1**@CNT(22.1) has none PXRD diffraction of supramolecular compound due possibly to the low mass content of **1** in this composite. Such a hypothesis is supported by a typical diffraction signal of **1** at 5.1° on the composite with the increased mass content of phthalocyanine compound up to 75.0 wt% (Fig. S8). Energy dispersive X-ray spectroscopy (EDX) element mapping observes C, N, F, and Ni elements (Fig. S9), determining successful immobilization of **1** on CNTs. Comparative studies of the electronic structure of NiPc(OH)₆(DCNFO) in **1** and **1**@CNT were carried out on X-ray photoelectron spectroscopy (XPS) (Fig. 2d). In **1**, the binding energy of Ni 2p_{3/2} peak appears at 855.4 eV. After the deposition of **1** on CNTs, a slight shift of Ni 2p_{3/2} peak moves to 855.7 eV in the XPS spectrum of **1**@CNT, suggesting the strong π-π interactions between metallic phthalocyanine macrocycle and CNTs surface. Extended X-ray absorption fine structure (EXAFS) spectrum of the Ni K-edge for **1**@CNT(22.1) is similar to that of **1** (Fig. 2e and S10, Table S4), manifesting their similar Ni coordination geometries. It is worth noting that there is a slightly negative shift for a second intense peak at 8335.84 eV, which originates from the characteristic transition for Ni–N₄ coordination interactions, in the Ni K-edge X-ray absorption near-edge spectra (XANES) of **1**@CNT(22.1) relative to that of **1** (Fig. 2f). This evidence further discloses the presence of π-π interactions between **1** and CNTs [61,62,80,81].

3.4. Electrocatalytic CO₂RR in H-type cell

The electrocatalytic performances of **1**@CNT(*n*) (*n* = 7.8, 14.5, 22.1, and 29.8) in the H-type cell were investigated in comparison with that of **1**@CNT, **1**@CB (CB: conductive black) under the same conditions. **1**@CNT and **1**@CB were made by grinding the mixtures of **1** and different

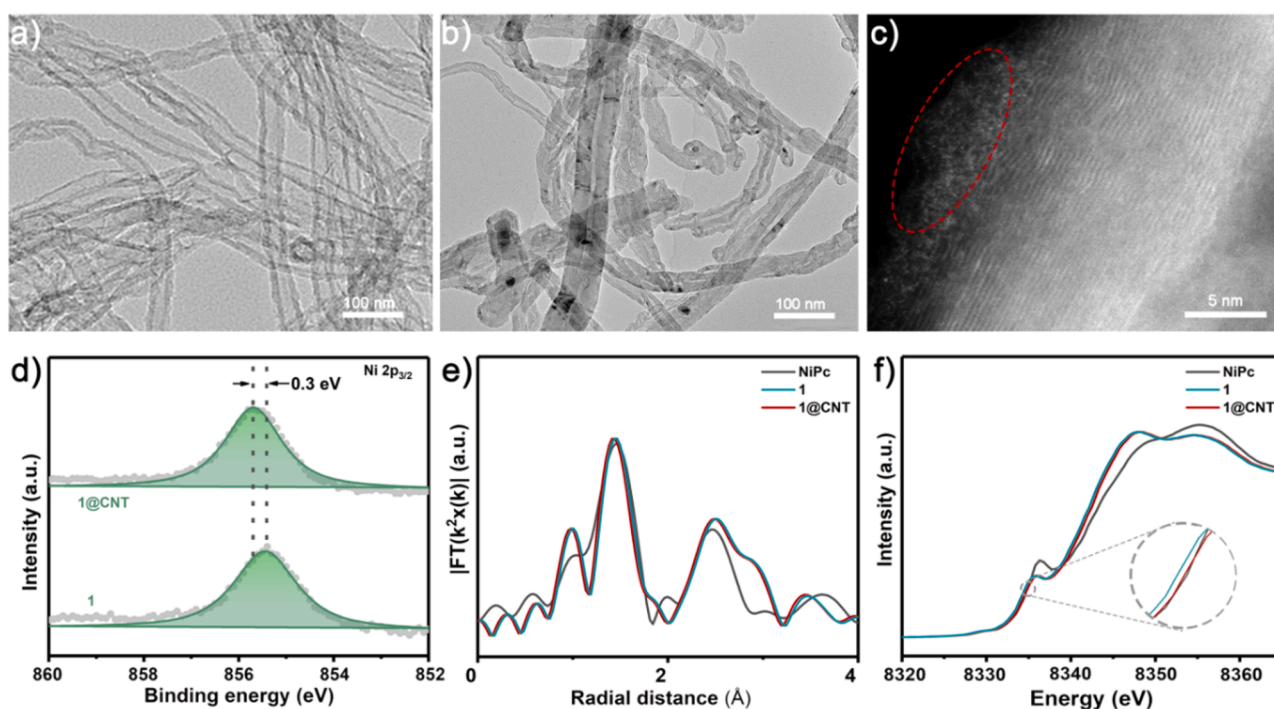


Fig. 2. (a) TEM image of CNTs; (b) TEM image of **1**@CNT(22.1); (c) AC HAADF-STEM photo of **1**@CNT(22.1); (d) Ni 2p_{3/2} XPS spectra of **1** and **1**@CNT(22.1); (e) Experimental curves of unsubstituted phthalocyanine (NiPc), **1**, and **1**@CNT; (f) Ni K-edge of XANES spectra of NiPc, **1**, and **1**@CNT.

conductive substances, respectively. The weight percentage of 22.1 wt% for **1** in **1@CNT** and **1@CB** was selected due to **1@CNT(22.1)** with the same active catalyst weight percentage showing the most excellent electrochemical catalytic activity among four *in-situ* prepared composites. The reference $\text{NiPc}(\text{OH})_8$ @CNT was also prepared by adding CNTs into the solvothermal solution containing only $\text{NiPc}(\text{OH})_8$. Gas chromatographic (GC) and liquid ^1H NMR spectroscopic results disclose the presence of only binary gas products including carbon monoxide (CO) and hydrogen (H_2) (Figs. S11 and S12). All mentioned potentials below were converted to the values versus RHE. **1@CNT(22.1)** electrode in Ar-saturated 0.5 M KHCO_3 electrolyte shows the very slow increase of current density along with the potential changing from 0 to 1.0 V in the linear sweep voltammetry (LSV) curve (Fig. 3a). In contrast, the replacement of Ar-saturated electrolyte with CO_2 -saturated species leads to the surge of current density, suggesting efficient CO_2 catalysis on **1@CNT(22.1)**. Moreover, LSV curves in CO_2 -saturated 0.5 M KHCO_3 electrolyte reveals that **1@CNT(22.1)** presents a lower onset potential (-0.44 V, where the reduction current density reaches 0.1 mA cm^{-2}) and a larger total current density (68.3 mA cm^{-2}) at -1.0 V compared to the others with different loading amounts, Fig. S13. The optimal weight percentage of **1** on composite was thus identified as 22.1 wt%. Nevertheless, the onset potential of -0.44 V for **1@CNT(22.1)** is a bit positive than those for the controls including **1@CNT** (-0.47 V), **1@CB** (-0.49 V), and $\text{NiPc}(\text{OH})_8$ @CNT (-0.48 V), confirming the catalytic activity and advantages of *in-situ* growth of **1** on CNTs. Such tendency was also monitored in their CO partial current densities (j_{CO}) at the potential range of -0.5 to -1.1 V (Figs. 3b and S14). In particular, **1@CNT(22.1)** possesses an exceptional selectivity towards CO ($\text{FE}_{\text{CO}} > 98\%$) with a current density of 12.2 mA cm^{-2} at -0.7 V and 55.1 mA cm^{-2} at -1.1 V, much higher than those of **1@CNT** (6.6 mA cm^{-2} at -0.7 V and 25.8 mA cm^{-2} at -1.1 V), **1@CB** (1.8 mA cm^{-2} at -0.7 V and 20.8 mA cm^{-2} at -1.1 V), and $\text{NiPc}(\text{OH})_8$ @CNT (1.8 mA cm^{-2} at -0.7 V and 24.1 mA cm^{-2} at -1.1 V) (Fig. 3b). Meanwhile, the FE_{CO} of **1@CNT(22.1)** maintains a level of above 98% in the potential range from -0.7 to -1.1 V, Fig. 3c, demonstrating an enhanced selectivity of CO for CO_2RR in comparison with **1@CNT**,

1@CB, and $\text{NiPc}(\text{OH})_8$ @CNT. These results further evidence the material advantages of **1@CNT(22.1)** in electrocatalysis. The nickel content of **1@CNT(22.1)** amounts to ca. 1.0 wt% based on ICP-AES test (Table S5). As a result, the TOF is calculated as 3.3 s^{-1} at -1.1 V, higher than those of all thus far reported nickel phthalocyanine-based polymeric electrocatalysts [45,46,75,78] and some nickel-based single-atom catalysts (SACs) [82–87], (Fig. 3d and Table S6). It is worth noting that j_{CO} of **1@CNT(22.1)** with current density of -49 mA cm^{-2} at -1.0 V is much bigger than those of most state-of-the-art molecule-based electrodes with polymeric structures such as NiPc-NiO_4 (24 mA cm^{-2} at -1.0 V) [75] and NiPc-TFPNCOF (15 mA cm^{-2} at -1.0 V) [46] under the similar operation conditions, revealing the excellent catalytic activity of **1@CNT(22.1)** (Fig. 3e).

Tafel slope of 75 mV dec^{-1} for **1@CNT(22.1)** is much smaller than those for **1@CNT** (94 mV dec^{-1}) and **1@CB** (113 mV dec^{-1}), demonstrating its faster dynamic process towards the electrochemical CO_2RR due to the thin-layered dispersion of **1** on CNTs prepared by the *in-situ* preparation method (Fig. S16). In addition, electrochemical impedance spectroscopy (EIS) data reveal that the charge transfer resistances of the composite electrocatalysts rank in the order of **1@CNT(22.1)** < **1@CNT** < **1@CB** (Fig. S17). The smaller resistance for **1@CNT(22.1)** is beneficial to the efficient charge transfer from conductive substrates to metal phthalocyanine electrocatalyst. **1@CNT(22.1)** also displays considerable electrochemical stability, as indicated by the retained 98% initial j_{CO} with constant FE_{CO} of 98% after 10 h cycling operation at -0.9 V (Fig. 3f).

3.5. Electrocatalytic CO_2RR in flow cell

A series of above-mentioned electrocatalysts including **1@CNT(n)**, **1@CNT**, **1@CB**, and $\text{NiPc}(\text{OH})_8$ @CNT electrodes were installed as gas diffusion electrode on a flow cell as the CO_2RR electrolyzer (please see Fig. S19 in Supplementary Material for the device principle figure and methods for details). In 0.5 M KHCO_3 electrolyte, the current densities for these electrodes in flow cell are much superior to corresponding values determined in H-type cell. As shown in Fig. 4a, **1@CNT(22.1)**

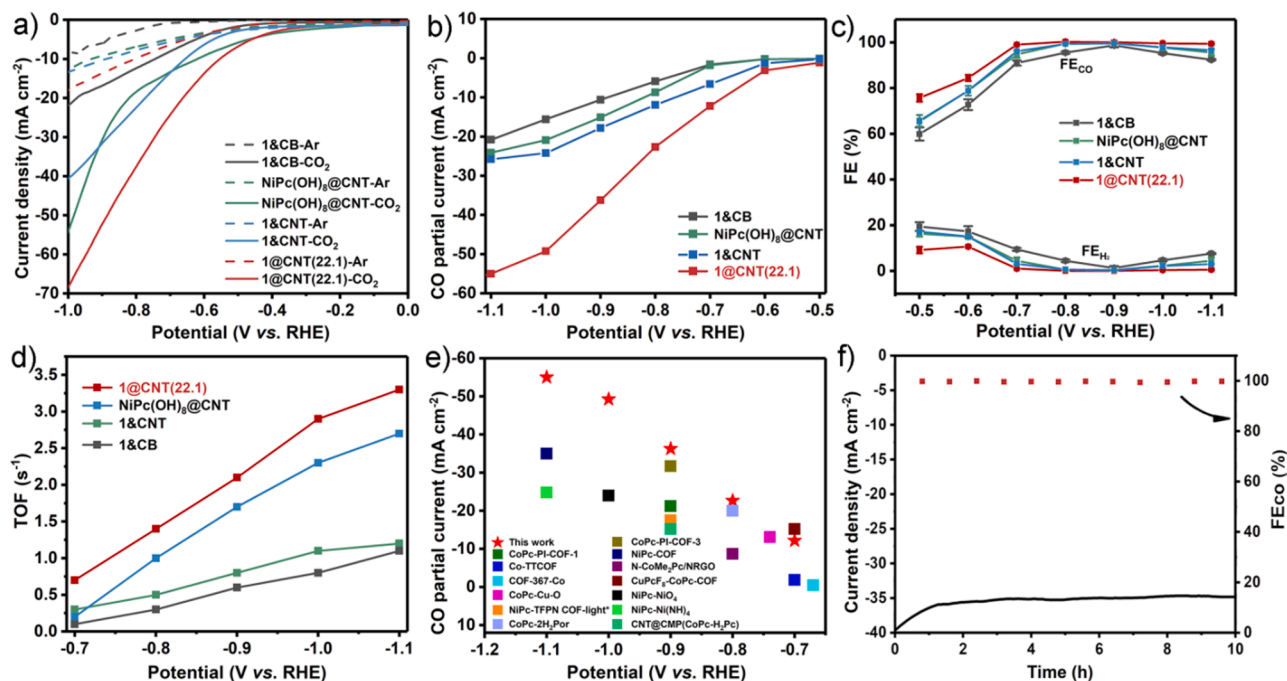


Fig. 3. (a) LSV profiles of **1@CNT(22.1)**, **1@CB**, $\text{NiPc}(\text{OH})_8$ @CNT, and **1@CNT** in Ar- and CO_2 -saturated 0.5 M KHCO_3 electrolytes at a scan rate of 10 mV s^{-1} ; (b) Current densities of **1@CNT(22.1)**, **1@CB**, $\text{NiPc}(\text{OH})_8$ @CNT, and **1@CNT** at different potentials in 0.5 M KHCO_3 electrolyte; (c) FE_{CO} and FE_{H_2} of **1@CNT(22.1)**, **1@CB**, $\text{NiPc}(\text{OH})_8$ @CNT, and **1@CNT** in a H-type cell; (d) TOF of **1@CNT(22.1)**, **1@CB**, $\text{NiPc}(\text{OH})_8$ @CNT, and **1@CNT** in a H-type cell; (e) Comparison of j_{CO} of **1@CNT(22.1)** electrocatalyst performed in H-type electrochemical cell; (f) Stability of **1@CNT** at a potential of -0.9 V versus RHE in a H-type cell.

possesses a current density of 145 and 312 mA cm⁻² at -0.9 and -1.1 V, respectively, being 3.0- and 4.7-time larger than the corresponding values of 36 and 55 mA cm⁻² measured in H-type cell. Such excellent electrochemical activity of 1@CNT(22.1) is originated from fast CO₂ diffusion to active electrocatalyst in gas diffusion electrode [39]. In the flow cell upon the employment of these electrocatalysts, still only H₂ and CO were detected by GC. FE_{CO} of 1@CNT(22.1) was able to be constantly kept above 98% at a whole current scope up to 353 mA cm⁻² (Fig. 4a and b), representing one of the best electrocatalytic activities among the molecule-based supramolecular catalysts and outperforming many excellent covalent-linked materials including NiPc-COF [45], NiPcP [49], and NiPPc/CNT [51], and even comparable with those in other SACs, noble metal and molecule electrocatalysts [88–95], Table S7. In contrast, 1@CNT, 1@CB, and NiPc(OH)₈@CNT exhibit the high FE_{CO} above 98% with a much narrower voltage ranges, Fig. 4b, further indicating the materials advantage of in-situ immobilization of supramolecular electrocatalysts on conductive substrate in electrocatalysis. Due to the intrinsic flooding issues of the flow cell, cycling test of 1@CNT(22.1) was checked at the current of 150 mA cm⁻² for 40 h (Fig. 4c). The slight decay of current density was detected with the FE_{CO} maintained above 98%, revealing the superior stability of this composite in electrochemical CO₂ conversion.

Interestingly, replacing 0.5 M KHCO₃ electrolyte by 1.0 M KOH in the flow cell led to an increased j_{CO} of 1@CNT(22.1) to 620 mA cm⁻² at -1.2 V with a high FE_{CO} of 98% (Fig. S20). Subsequently, the constant-current electrolysis of 1@CNT(22.1) at the current density of 230 mA cm⁻² was performed. As shown in Fig. 4d, no significant increase in the applied potential is noticed during 40 h continuous test, further confirming the excellent durability of 1@CNT(n) catalyst in this flow cell device. Herein, it is noteworthy that the present work reports the effective immobilization of molecule-based catalysts on the

electrode by solvothermal formation of insoluble supramolecular polymer on conductive substrate through π - π interaction. It is different from the heterogeneous methodology through sole physical adsorption, covalent linking, and polymerization, but similar to that for organic frameworks materials [24]. In addition, this work is able to modify microenvironment of active site by simultaneous introduction of substituents and immobilization on carbon tubes, it also enhances the catalyst stability by forming a new supramolecular polymer.

4. Theoretical calculations

In order to rationalize the excellent electrocatalytic behavior of 1@CNT(22.1), density functional theory (DFT) calculation for NiPc(OH)₆(DCNFO) was carried out. In comparison with NiPc(OH)₈, NiPc(OH)₆(DCNFO) has a broader conjugating structure constructed by the covalent fusion of two aromatic systems of NiPc(OH)₆ and DCNFO. As shown in Figs. S21 and S22, the π electron localized orbital locator (π -LOL) calculation data clearly show the bigger conjugating system for NiPc(OH)₆(DCNFO) relative to that of NiPc(OH)₈. The large conjugating system of NiPc(OH)₆(DCNFO) leads to a narrow HOMO-LUMO gap, which benefits to the electron transmission in the present molecular material [96], and thus the catalyzing activity. It is worth noting that charge differential density of NiPc(OH)₆(DCNFO)@C (representing a NiPc(OH)₆(DCNFO) molecule located on graphite segment instead of 1@CNT) also was explored according to DFT calculations toward understanding the interaction between NiPc(OH)₆(DCNFO) and CNTs in the electrocatalyst. As shown in Fig. S23, observation of strong inter-layered π - π interactions between NiPc(OH)₆(DCNFO) and carbon interface is consistent with the previous XPS test results, facilitating the rapid electron transport in the composite electrode [97–99].

The electron density difference (EDD) result proves that the electron

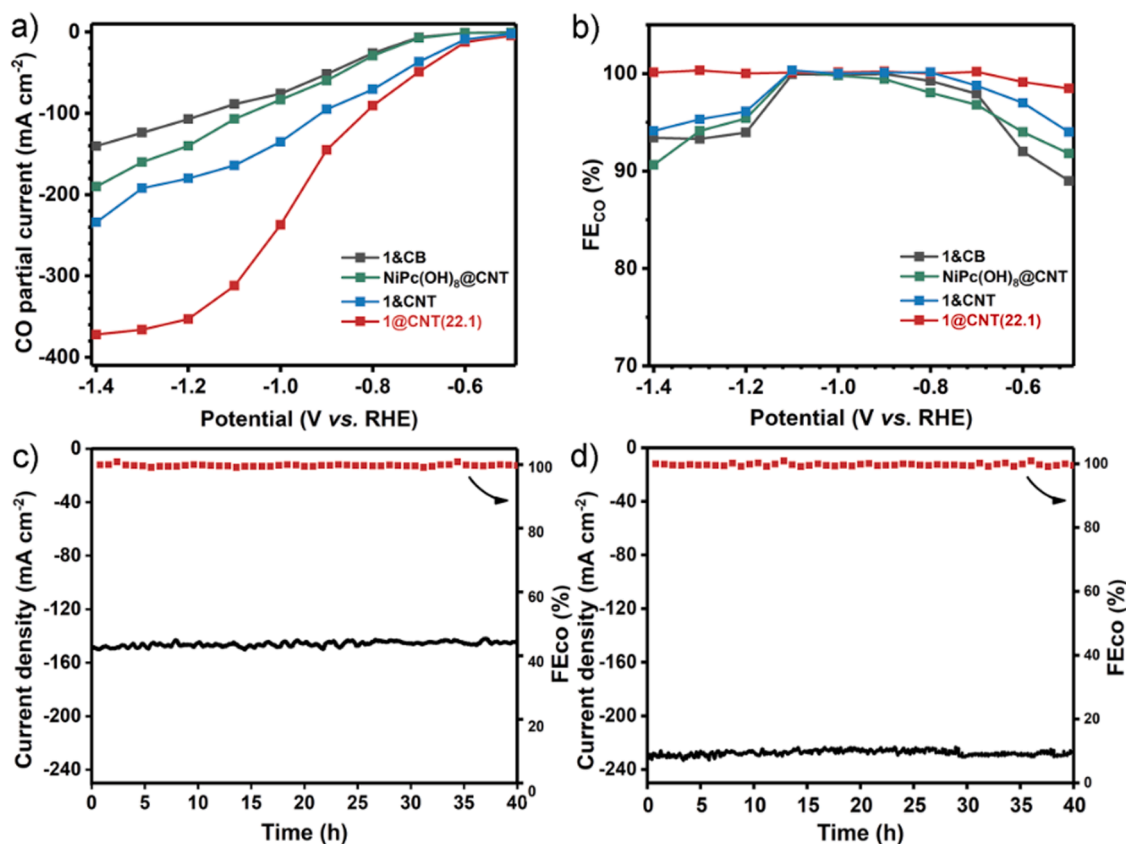


Fig. 4. (a) Current densities of 1@CB, NiPc(OH)₈@CNT, 1@CNT and 1@CNT(22.1) at different potentials in a flow cell with 0.5 M KHCO₃ electrolyte; (b) FE_{CO} of 1@CB, NiPc(OH)₈@CNT, 1@CNT and 1@CNT(22.1) in a flow cell with 0.5 M KHCO₃ electrolyte; (c) Stability of 1@CNT(22.1) at 150 mA cm⁻² in a flow cell with 0.5 M KHCO₃ electrolyte; (d) Stability of 1@CNT(22.1) at 230 mA cm⁻² in a flow cell with 1.0 M KOH electrolyte.

density is crowded around the Ni atom of metallic phthalocyanines (Fig. 4a). In addition, a charge density calculation further proves that the electron density of $+2.049 \text{ e } \text{\AA}^{-3}$ for Ni atom in $\text{NiPc}(\text{OH})_6(\text{DCNFO})$, higher than that in $\text{NiPc}(\text{OH})_8$ ($+2.040 \text{ e } \text{\AA}^{-3}$) and NiPc ($+2.032 \text{ e } \text{\AA}^{-3}$). This result is in good agreement with the Ni K-edge XANES data. The higher electron concentration around the Ni atom should be helpful to absorb the CO_2 molecule (Figs. 5a, 5b and S24). In addition, the Gibbs free energy diagram for the CO_2RR pathways was also calculated on $\text{NiPc}(\text{OH})_6(\text{DCNFO})$, $\text{NiPc}(\text{OH})_8$, and NiPc , respectively (Fig. 5e). According to the calculated electrochemical reducing diagram for the $\text{CO}_2 \rightarrow \text{CO}$ conversion, a standard proton-coupled electron transfer (PCET), $* + \text{CO}_2 + \text{H}^+ + \text{e}^- \rightarrow * \text{COOH}$, takes place initially as the rate-determining step (R.D.S.) for all the three model catalysts [100,101]. Among the present three model catalysts, $\text{NiPc}(\text{OH})_6(\text{DCNFO})$ has the lowest energy barrier of 0.62 eV, indicating its highest catalyzing efficiency.

The molecular structural calculation on the X-[NiY] complexes [X = CO_2 or CO ; Y = $\text{Pc}(\text{OH})_6(\text{DCNFO})$, $\text{Pc}(\text{OH})_8$, or Pc] also provides useful information to understand the excellent catalyzing efficiency of $\text{Pc}(\text{OH})_6(\text{DCNFO})$. Before the reaction, the distance between CO_2 carbon atom and Ni atom of molecular electrocatalyst is 2.87, 2.95, and 3.02 Å for $\text{NiPc}(\text{OH})_6(\text{DCNFO})$, $\text{NiPc}(\text{OH})_8$, and NiPc , respectively, indicating the strongest interaction between CO_2 and the former compound. For the desorption process of electrocatalytic CO_2RR , the distance between CO carbon atom and Ni atom from electrocatalyst is 2.84, 2.82, and 2.81 Å for $\text{NiPc}(\text{OH})_6(\text{DCNFO})$, $\text{NiPc}(\text{OH})_8$, and NiPc , respectively, suggesting the weakest attraction between CO and $\text{NiPc}(\text{OH})_6(\text{DCNFO})$

(Fig. S25). As a result, the strongest CO_2 adsorption and CO desorption abilities of $\text{NiPc}(\text{OH})_6(\text{DCNFO})$ among three species are helpful for corresponding CO_2RR electrocatalysis.

5. Conclusion

In summary, a supramolecular structure based a new A_3B -type asymmetric phthalocyanine nickel molecule directly prepared by the solvothermal reaction has been introduced, displaying the comparable solvent resistance to MOFs. This supramolecular structure has been used as a molecule-based polymeric electrocatalyst, rather than a dispersed molecular sample, towards carbon dioxide reduction reaction. In comparison with control electrodes made up of different molecular catalyst and various conductive substrates prepared by mechanical grinding, a composite generated by in-situ growth of this compound on carbon tubes exhibits superior electrocatalytic activities to selectively convert CO_2 to CO under industry-compatible current densities. The excellent electrocatalytic activity of this composite electrode has been well correlated from both experimental and theoretical perspectives. The present work not only provides a new excellent molecule-based polymeric electrocatalyst to provide industry-compatible current density and excellent selectivity, but also could give a clear hint that the electrocatalytic activities of MOFs and COFs electrodes reported previously might be further improved by using a suitable fabrication method to disperse active materials on conductive substrates.

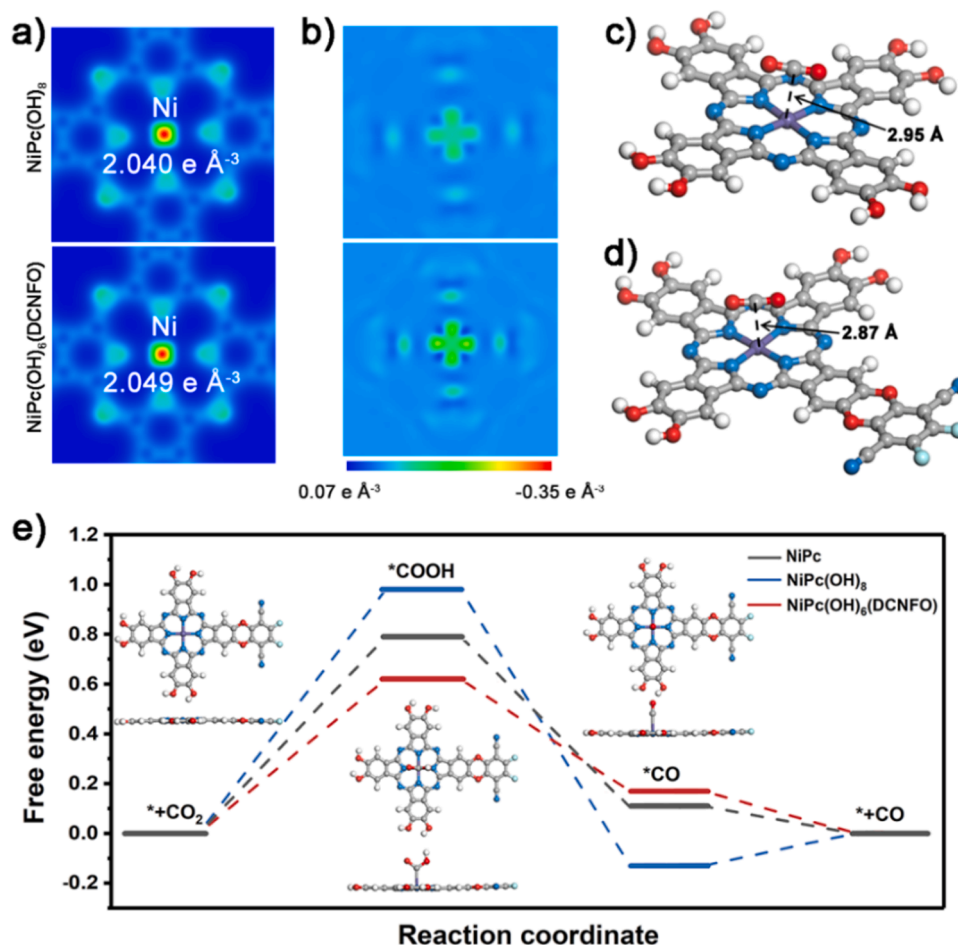


Fig. 5. (a) Charge density differences (the cyan and orange indicate electron accumulation and depletion) of $\text{NiPc}(\text{OH})_8$ and $\text{NiPc}(\text{OH})_6(\text{DCNFO})$; (b) Electron densities of $\text{NiPc}(\text{OH})_8$ and $\text{NiPc}(\text{OH})_6(\text{DCNFO})$; (c, d) Molecular structures of CO_2 -adsorbed $\text{NiPc}(\text{OH})_8$ and $\text{NiPc}(\text{OH})_6(\text{DCNFO})$; (e) Free energy diagrams of $\text{NiPc}(\text{OH})_8$ and $\text{NiPc}(\text{OH})_6(\text{DCNFO})$ for CO_2RR (inset: molecular structure of $\text{NiPc}(\text{OH})_6(\text{DCNFO})$ in CO_2RR process).

CRediT authorship contribution statement

Y.J. and Y.Z. conceived the project and designed the experiments. X. Z. and H.W. fabricated the electrocatalytic measurements and analysed the data. X.L. and B.Y. performed SEM and TEM characterizations. X.D. and T.Z. performed theoretical computations. Y.J. and H.W. wrote the manuscript. K.W., D.Q. and J.J. supervised the work and polished the manuscript.

Declaration of Competing Interest

The authors declare that they have no known competing financial interests or personal relationships that could have appeared to influence the work reported in this paper.

Data availability

Data will be made available on request.

Acknowledgments

This work was financially supported by the Natural Science Foundation of China (Nos. 22235001, 22131005, 22011540002, 22175020, 21871024, and 21631003), Xiaomi Young Scholar Programme, the Fundamental Research Funds for the Central Universities, and University of Science and Technology Beijing.

Appendix A. Supplementary material

Supplementary data associated with this article can be found in the online version at [doi:10.1016/j.apcatb.2023.122446](https://doi.org/10.1016/j.apcatb.2023.122446).

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